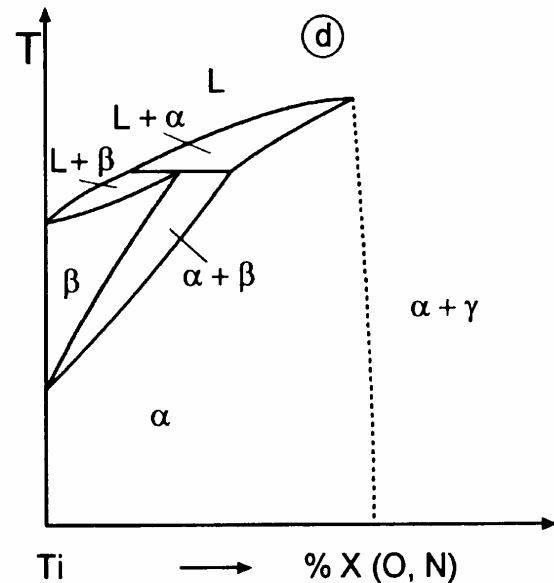
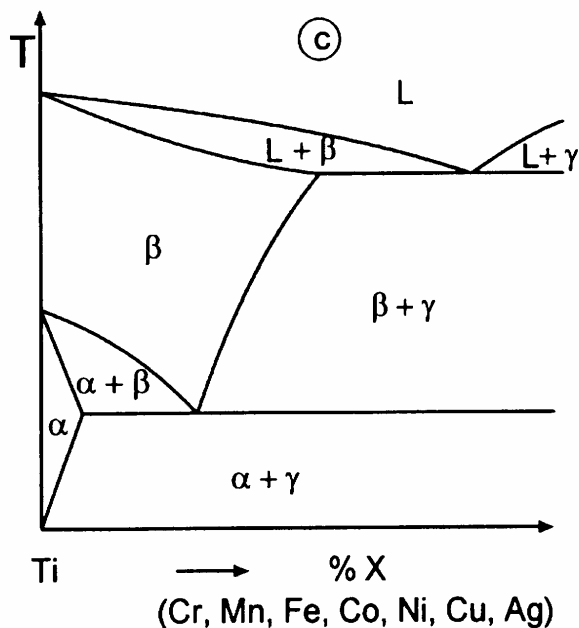
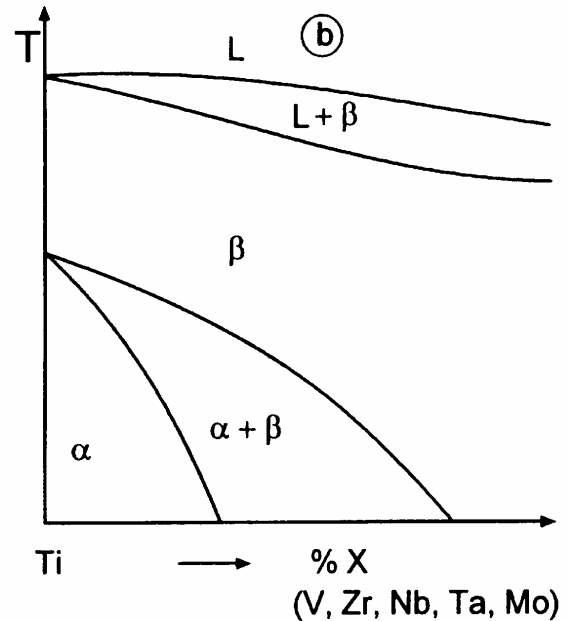
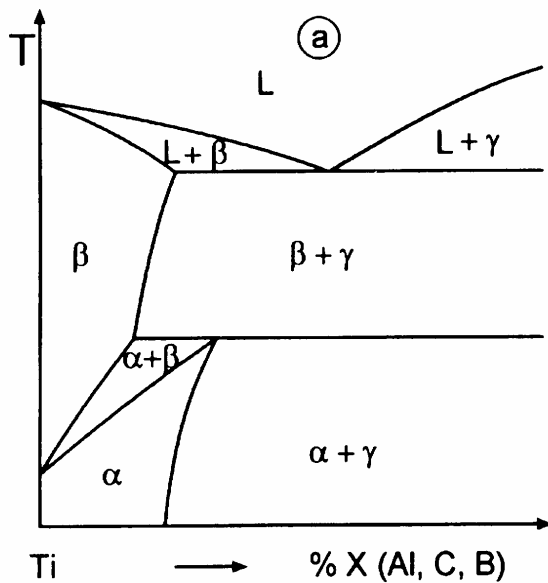


Titanium based materials / Combinations of metallic biomaterials / Special Materials

8.5. Titanium based materials

8.5.1. Groups of alloys



α -alloys (hexagonal Closed packed) β -alloys (body Centered Cubic)

Effects of alloying elements on titanium alloy structure

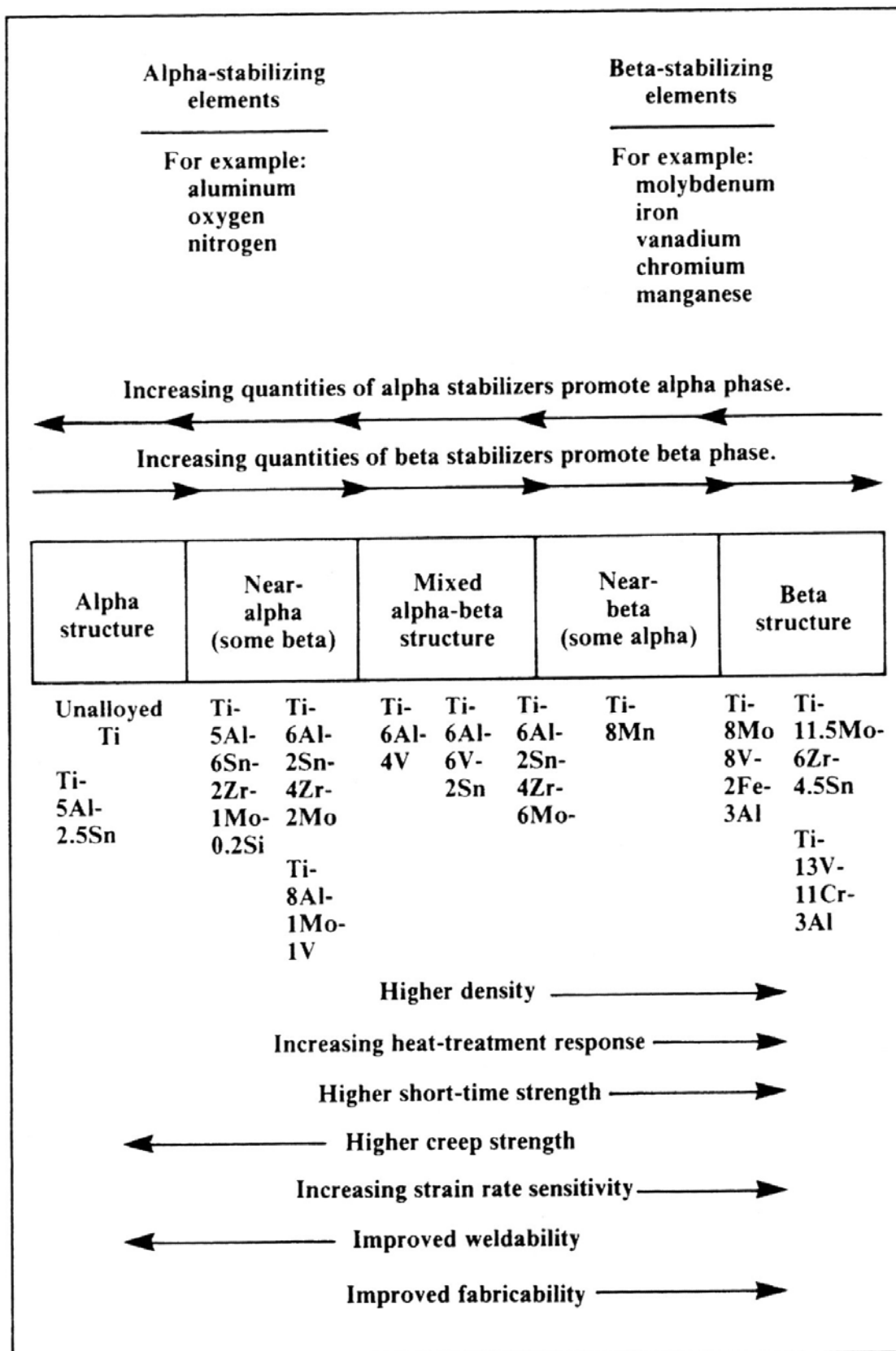


Fig. 3.5. Effects of alloying elements on titanium alloy structure. (Reproduced by permission of ASM International, Metals Park OH, USA).

Chemical compositions of titanium alloys

Table 3.1. Chemical composition and standards for selected titanium biomaterials.

Grade Name and Type	UNS Number	ASTM Standard	ISO Standard	Chemical Composition Nominal Weight %
Ti CP-1 (Alpha)	R50250	ASTM F 67	ISO 5832-2	C 0.10 max Fe 0.20 max H 0.015 max N 0.03 max O 0.18 max Ti rem
Ti CP-2 (Alpha)	R50400	ASTM F 67	ISO 5832-2	C 0.10 max Fe 0.30 max H 0.015 max N 0.03 max O 0.25 max Ti rem
Ti CP-3 (Alpha)	R50550	ASTM F 67	ISO 5832-2	C 0.10 max Fe 0.30 max H 0.015 max N 0.05 max O 0.35 max Ti rem
Ti CP-4 (Alpha)	R50700	ASTM F 67	ISO 5832-2	C 0.10 max Fe 0.50 max H 0.015 max N 0.05 max O 0.40 max Ti rem
Ti-3Al-2.5V (Alpha/Beta)	R56320	ASTM B 348 F-04.12.06	---	Al 3.0 C 0.05 max Fe 0.13 H 0.015 max N 0.01 O 0.10 Ti rem V 2.5
Ti-5Al-2.5Fe (Alpha/Beta)	---	---	ISO 5832-10	Al 5.0 C 0.05 max Fe 2.5 H 0.015 max N 0.05 max O 0.25 Ti rem
Ti-6Al-4V (Alpha/Beta)	R56400	ASTM F 1472	ISO 5832-3	Al 6.0 C 0.10 max Fe 0.20 H 0.015 max N 0.03 O 0.20 max Ti rem V 4.0
Ti-6Al-4V ELI (Alpha/Beta)	R56401	ASTM F 136	ISO 5832-3	Al 6.0 C 0.10 max Fe 0.20 H 0.015 max N 0.03 O 0.13 max Ti rem V 4.0
Ti-6Al-7Nb (Alpha/Beta)	R56700	ASTM F 1295	ISO 5832-11	Al 6.0 C 0.08 max Fe 0.15 H 0.009 max Nb 7.0 N 0.03 O 0.20 max Ta 0.5 max Ti rem
Ti-15Mo (Beta)	R58150	ASTM F 2066	---	C 0.05 max Fe 0.1 H 0.015 max Mo 15.0 N 0.01 O 0.15 Ti rem
Ti-13Nb-13Zr (Beta)	R58130	ASTM F 1713	---	C 0.08 max Fe 0.1 H 0.02 max Nb 13.0 N 0.01 O 0.1 Ti rem Zr 13.0
Ti-16Nb-10Hf (Beta)	---	---	---	C 0.05 max Fe 0.05 H 0.015 max Hf 9.5 Nb 16.0 N 0.002 O 0.1 Ti rem
Ti-15Mo-2.8Nb-0.2Si (Beta)	---	---	---	C 0.02 max Fe 0.2 H 0.02 max Mo 15.0 N 0.01 O 0.18 Ti rem Nb 2.8 Si 0.2
Ti-12Mo-6Zr-2Fe (Beta)	R58120	ASTM F 1813	---	C 0.02 max Fe 2.0 H 0.02 max Mo 12.0 N 0.01 O 0.18 Ti rem Zr 6.0
Ti-12Mo-5Zr-5Sn (Beta)	---	---	---	C 0.02 max Fe 0.2 H 0.02 max Mo 12.0 N 0.01 O 0.18 Ti rem Zr 5.0 Sn 5.0
Ti-15Mo-5Zr-3Al (Beta)	---	---	---	Al 3.0 C 0.02 max Fe 0.2 H 0.02 max Mo 15.0 N 0.01 O 0.18 Ti rem Zr 5.0
Ti-30Ta (Beta)	---	---	---	C 0.05 max Fe 0.1 H 0.015 max N 0.01 O 0.15 Ti rem Ta 30.0
Ti-45Nb (Beta)	R58450	AMS 4982	---	C 0.04 max Fe 0.1 H 0.003 max Nb 45.5 N 0.01 O 0.1 Ti rem
Ti-35Zr-10Nb (Beta)	---	---	---	C 0.05 max Fe 0.05 H 0.015 max Nb 10.5 N 0.002 O 0.1 Ti rem Zr 35.2
Ti-35Nb-7Zr-5Ta (Beta)	R58350	Task Force F-04.12.23	---	C 0.05 max Fe 0.05 H 0.015 max Nb 35.5 N 0.02 O 0.15 Ta 5.7 Ti rem Zr 7.3
Ti-55.8Ni (Intermetallic)	---	ASTM F 2063	---	C 0.05 max Fe 0.02 H 0.003 max Ni 55.8 N 0.01 O 0.05 Ti rem

Mechanical properties of selected titanium alloys

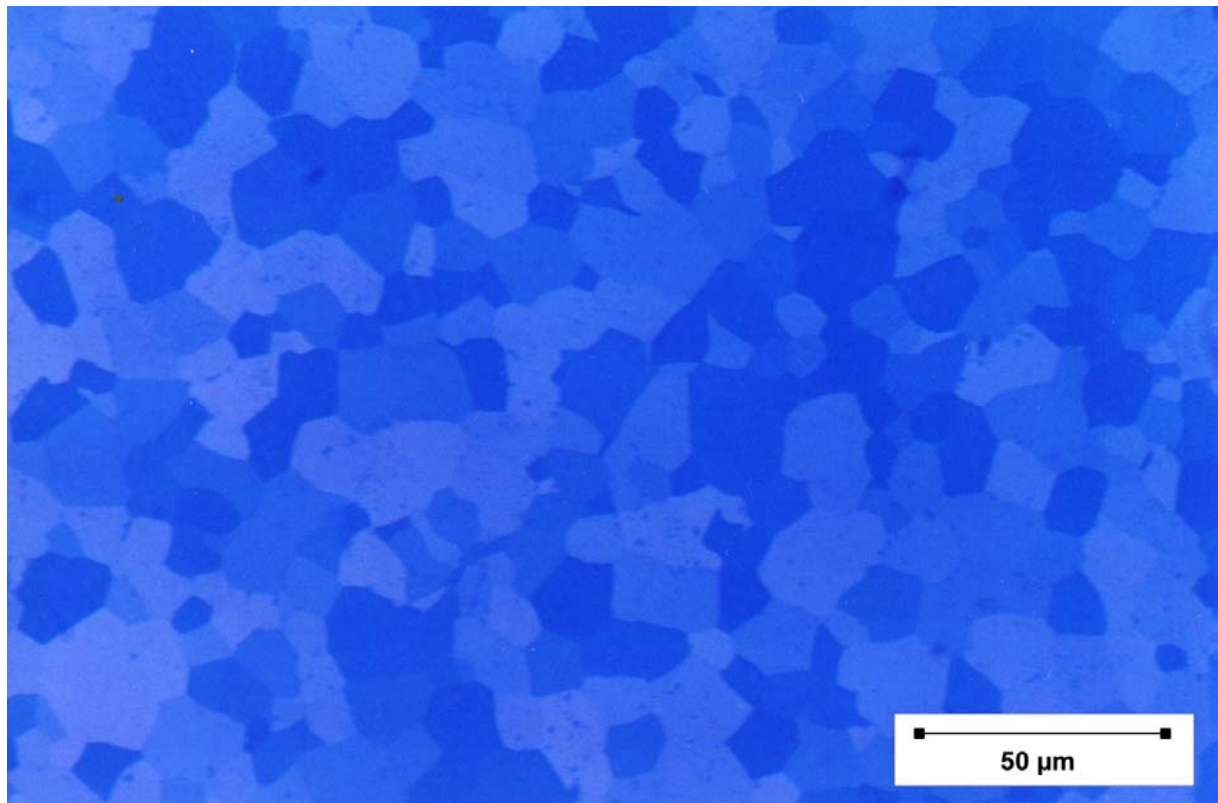
Table 3.2. Mechanical properties of selected titanium biomaterials.

Grade Designation and Type	Metallurgical Condition (footnote)	Tensile Strength ksi. (MPa)	0.2% Yield Strength ksi. (MPa)	Elongation %	Reduction in Area %	Typical Hardness (Rockwell)
Ti CP-1 (Alpha)	1,300 °F anneal (a)	35 (241)	25 (172)	24	30	70 HRB
Ti CP-2 (Alpha)	1,300 °F anneal (a)	50 (345)	40 (276)	20	30	80 HRB
Ti CP-3 (Alpha)	1,300 °F anneal (a)	65 (448)	55 (379)	18	30	90 HRB
Ti CP-4 (Alpha)	1,300 °F anneal (a)	80 (552)	70 (483)	15	25	100 HRB
Ti-3Al-2.5V (Alpha/Beta)	1,300 °F anneal (a)	100 (690)	85 (586)	15	25	24 HRC
Ti-5Al-2.5Fe (Alpha/Beta)	---	---	---	---	---	---
Ti-6Al-4V (Alpha/Beta)	1,300 °F anneal (a)	135 (931)	125 (862)	15	30	36 HRC
Ti-6Al-4V ELI (Alpha/Beta)	1,725 °F anneal (b)	125 (862)	115 (793)	10	25	32 HRC
Ti-6Al-7Nb (Alpha/Beta)	1,300 °F anneal (a)	125 (862)	115 (793)	10	25	32 HRC
Ti-15Mo (Beta)	1,475 °F anneal (c)	115 (793)	95 (655)	22	60	24 HRC
Ti-13Nb-13Zr (Beta)	Capability aged	125 (860)	105 (725)	8	15	---
Ti-16Nb-10Hf (Beta)	1,560 °F + water quench (d)	85 (486)	40 (276)	16	---	---
Ti-15Mo-2.8Nb-0.2Si (Beta)	1,475 °F anneal (c)	115 (793)	95 (655)	22	60	24 HRC
Ti-12Mo-6Zr-2Fe (Beta)	1,400 °F anneal (e)	145 (1,000)	140 (965)	15	40	33 HRC
Ti-12Mo-5Zr-5Sn (Beta)	---	---	---	---	---	---
Ti-15Mo-5Zr-3Al (Beta)	---	---	---	---	---	---
Ti-30Ta (Beta)	---	---	---	---	---	---
Ti-45Nb (Beta)	As drawn wire	70 (483)	65 (448)	12	55	---
Ti-35Zr-10Nb (Beta)	Hot Rolled	130 (897)	90 (621)	16	---	---
Ti-35Nb-7Zr-5Ta (Beta)	1,300 °F anneal (a)	120 (827)	115 (793)	20	55	35 HRC
Ti-55.8Ni (Intermetallic)	1,475 °F anneal (c)	150 (1,034)	50 (345)	20	---	---

HRB = Hardness, Rockwell B Scale. *HRC* = Hardness, Rockwell C Scale.

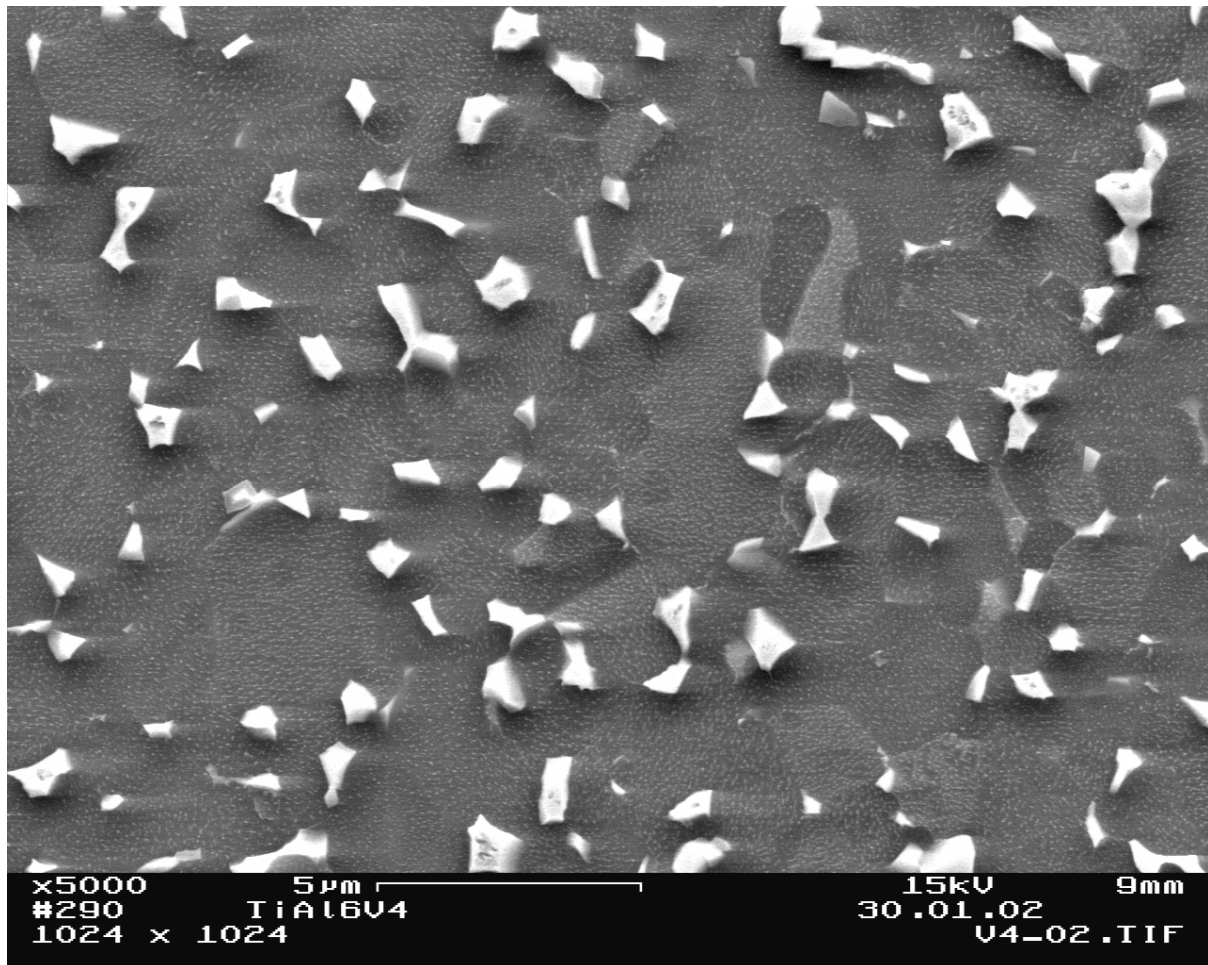
a: approx. 700 °C b: approx. 940 °C c: approx. 800 °C d: approx. 850 °C e: approx. 760 °C.

pure α -structure



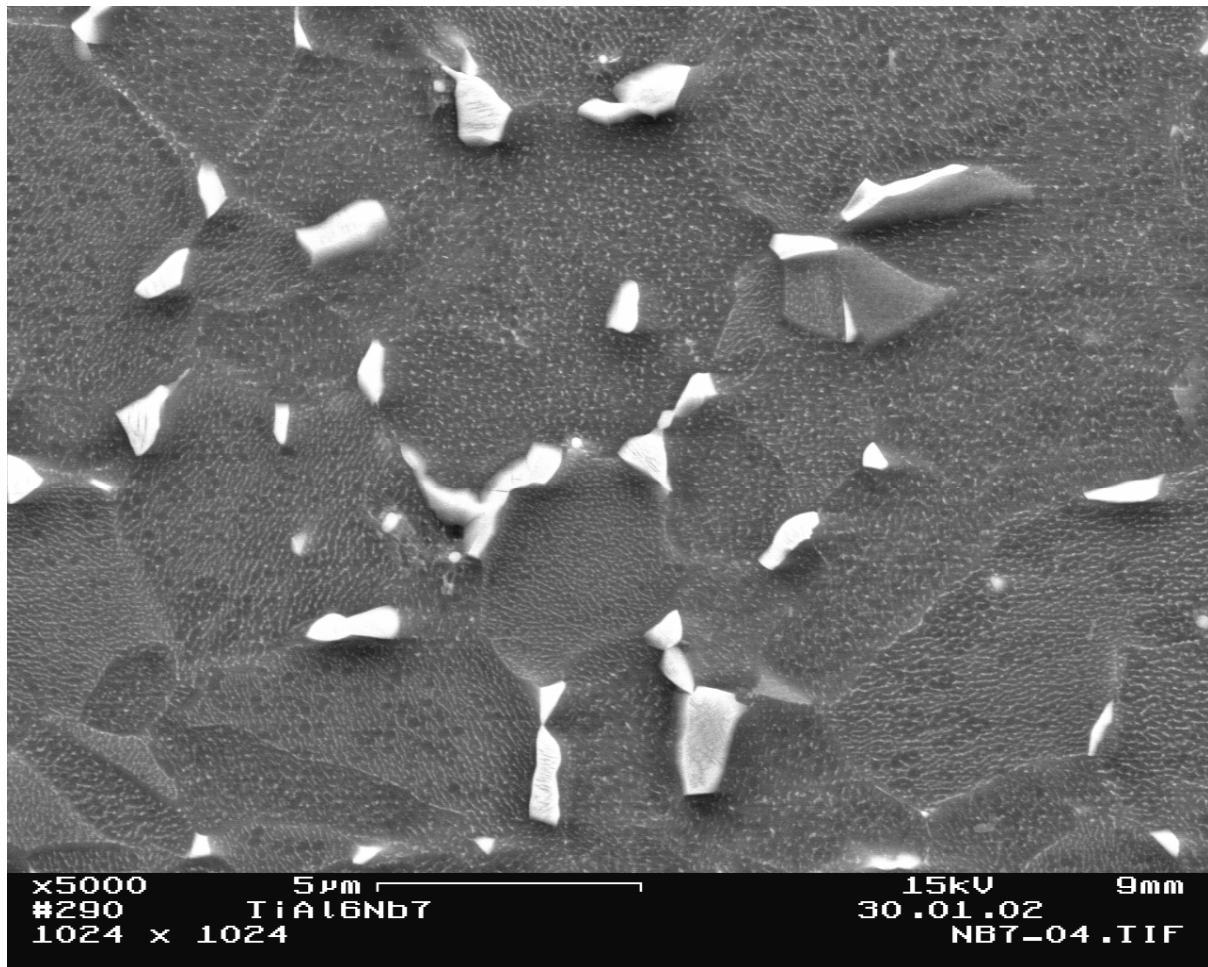
Ti, CP-1 ; average grain size = $10.8 \pm 5.6 \mu\text{m}$

Mixed α - β -structure



Ti6Al4V ; average grain size = $1.9 \pm 0.6 \mu\text{m}$
Dark = α ; bright = β (20 – 25 mass% V)

Mixed α - β -structure



Ti6Al7Nb ; average grain size = $3.2 \pm 0.6 \mu\text{m}$
Dark = α ; bright = β (18 – 21 mass% Nb)

Lateral distribution of alloying elements in Ti6Al7Nb

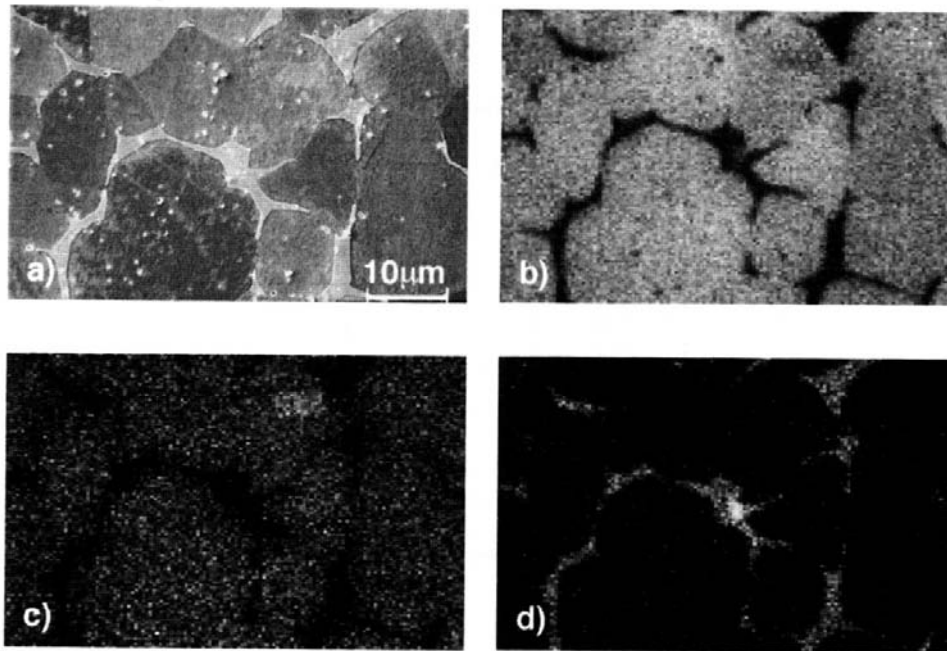


Fig. 7.17. Scanning Auger microscopy (SAM) images of the native oxide film on Ti-6Al-7Nb with artificially coarsened metal microstructure (to make imaging analysis easier): **a.** secondary electron image of the analyzed area; **b.** SAM map of Ti; **c.** Al; **d.** Nb [74]. (Reproduced from [74] by permission of Kluwer Publishers).

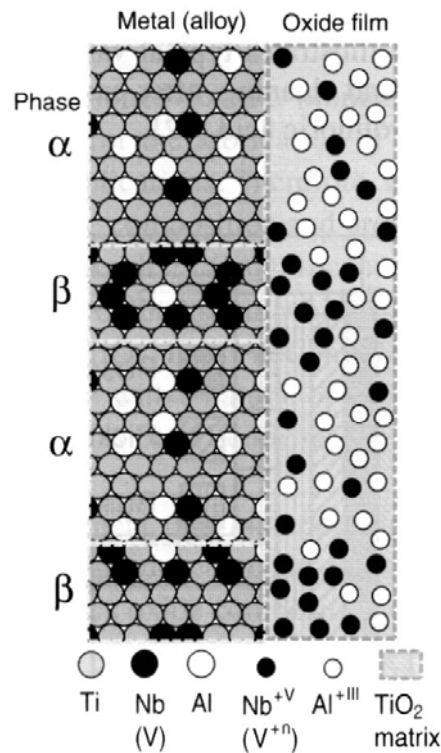


Fig. 7.15. Schematic representation (in cross-section) of a native oxide film on a heterogeneous alloy TiAlV or TiAlNb (for details: see text).

Influence of surface treatments on chemical surface composition

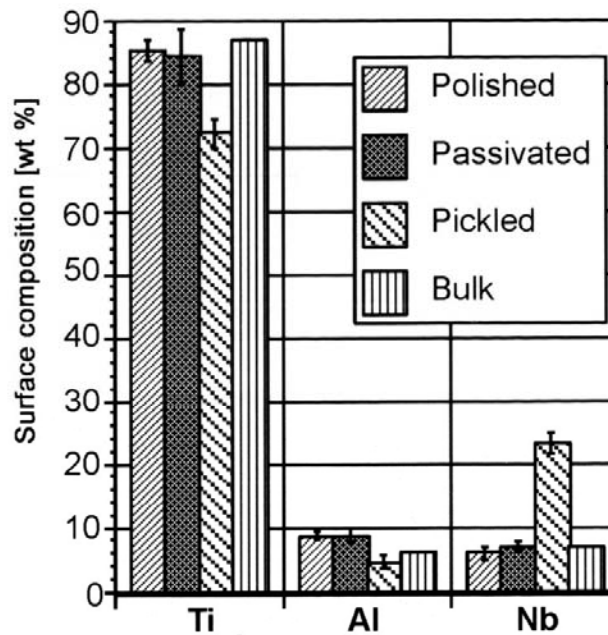
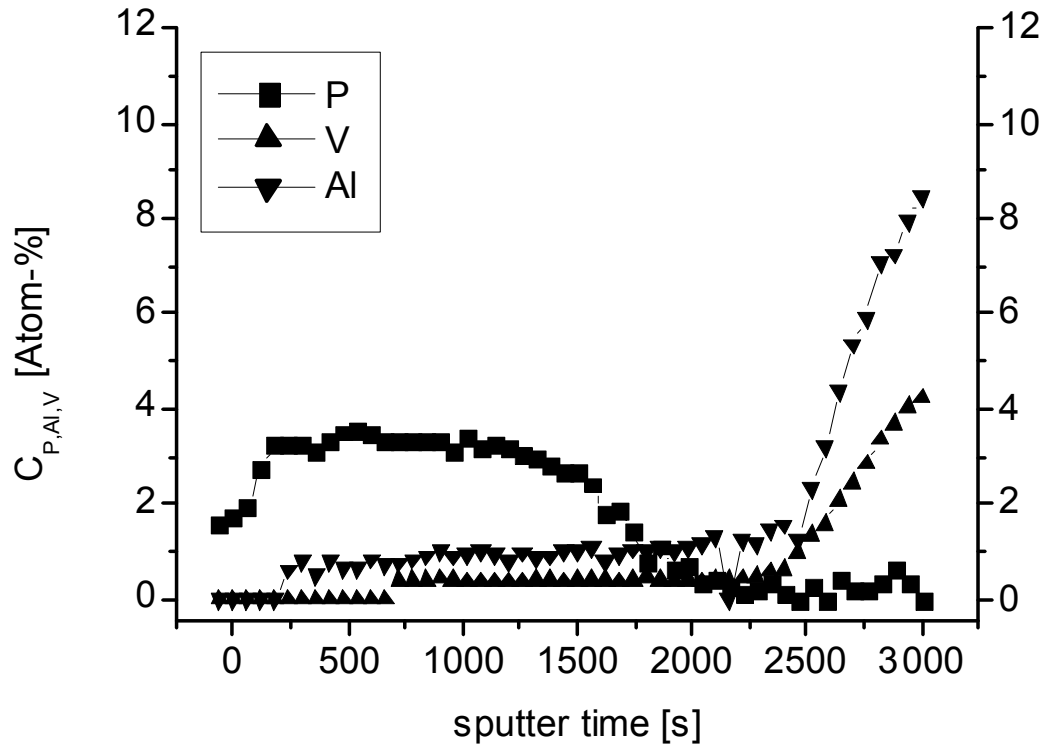


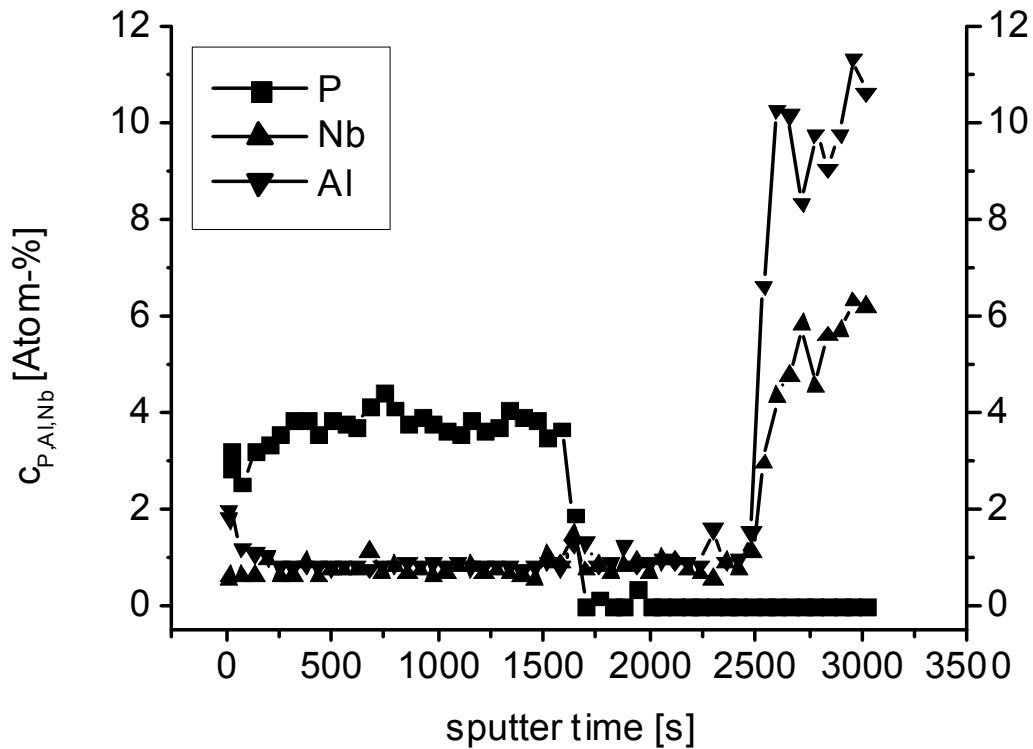
Fig. 7.16. Composition of the oxide film at the surface of alloy Ti-6Al-7Nb (XPS data converted to weight percent, normalized to Ti + Al + Nb = 100 wt%) compared with the bulk composition. The three different surfaces are mechanically polished and cleaned, passivated in nitric acid, and pickled in nitric acid/hydrofluoric acid respectively [74]. (Reproduced from [74] by permission of Kluwer Publishers).

Depth distribution of alloying elements in anodic oxide layers on Ti6Al4V



gs, Ti6Al4V; 6.37 mA cm^{-2} , 80 V
 $d_{\text{eff}} = 160 - 180 \text{ nm}$

Depth distribution of alloying elements in anodic oxide layers on Ti6Al7Nb



gs, Ti6Al7Nb; 6.37 mA cm^{-2} , 80 V
 $d_{\text{eff}} = 160 - 180 \text{ nm}$

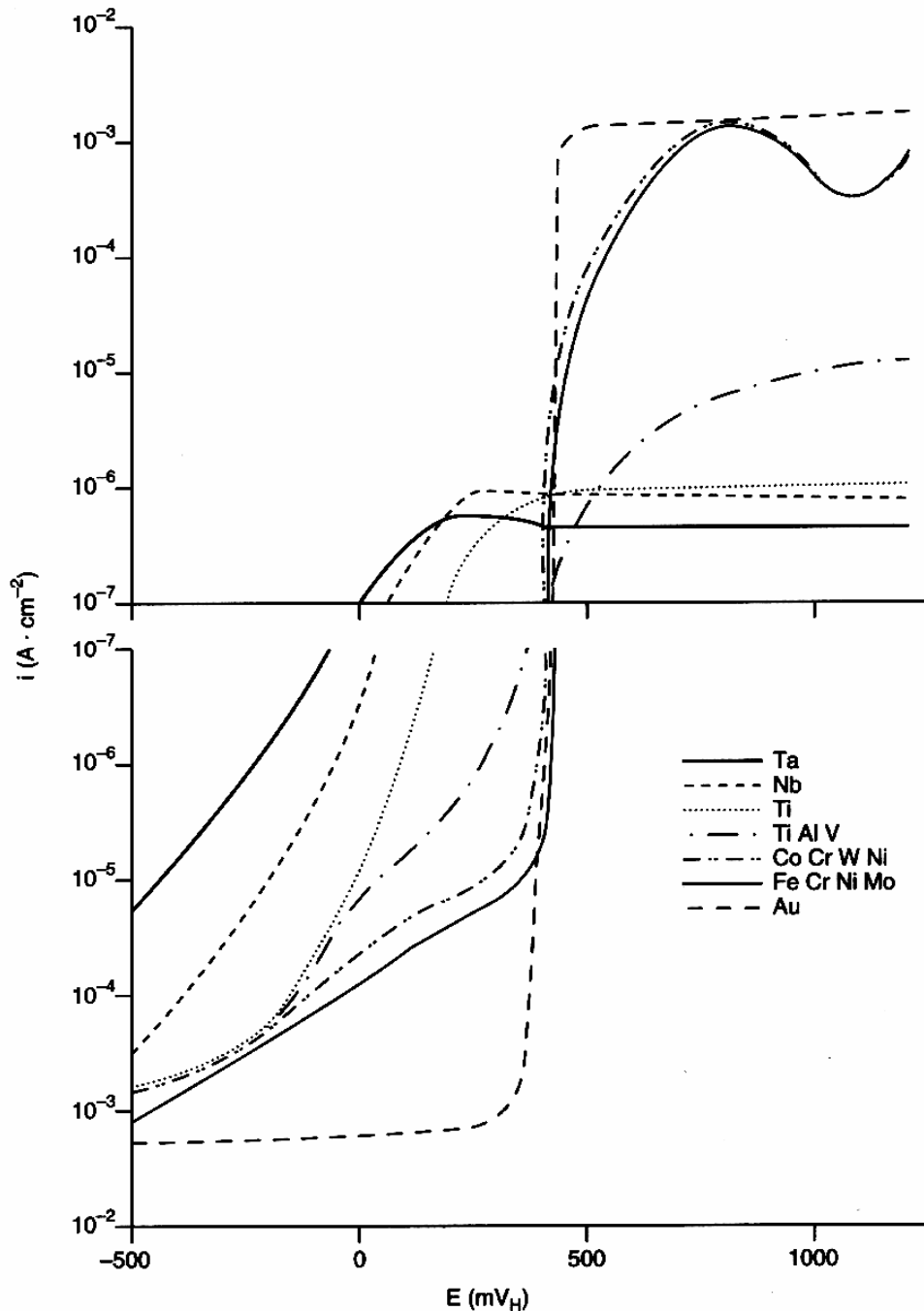
properties of oxide layers

conduction mechanism -

electronic

n-type semiconducting

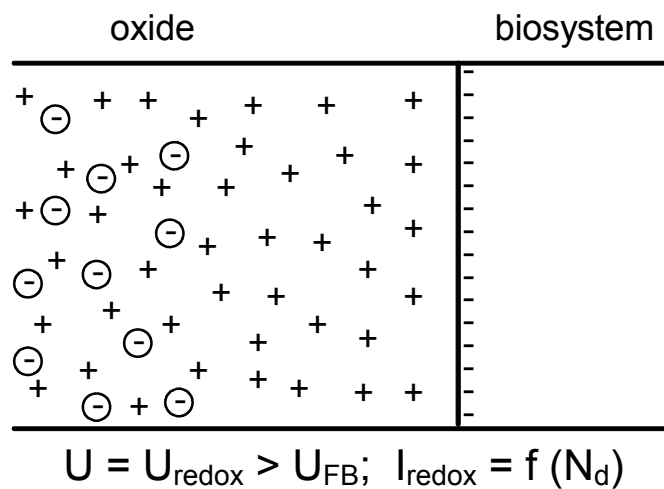
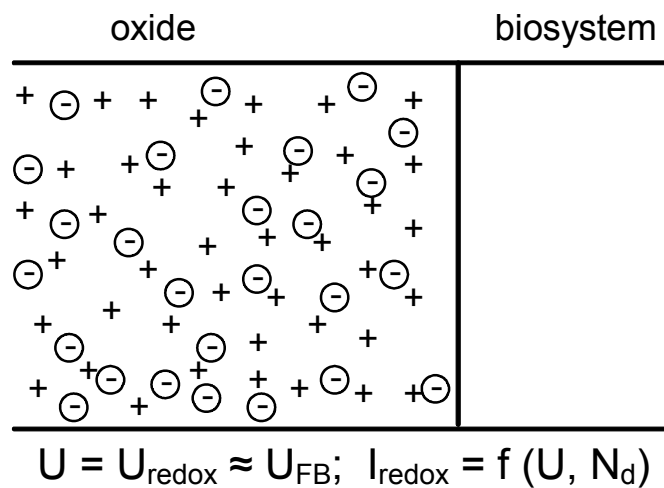
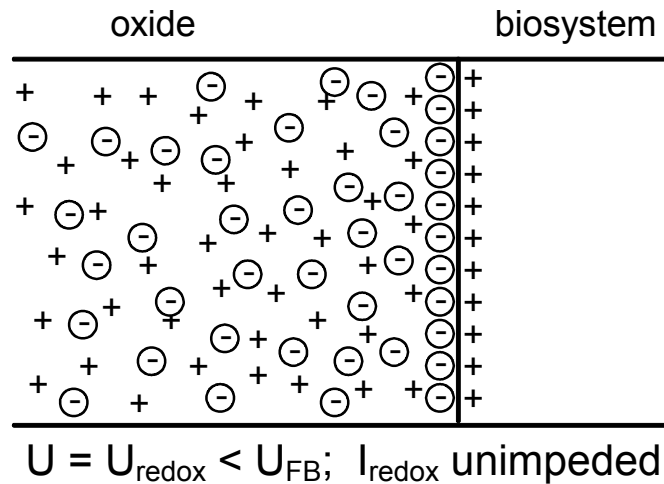
ionic



Oxide layers on titanium based alloys

n-type semiconductors

Characterized by flatband potential U_{FB} and donor density N_d



Mott-Schottky-Equation

$$\frac{1}{C_{sc}^2} = \frac{2}{q\epsilon\epsilon_0 N_d} \left(U - U_{FB} - \frac{kT}{q} \right) \approx \frac{1}{C^2}$$

with

C ... capacity

C_{SC} ... capacity of space charge layer in the n-type semiconductor

q ... charge of electron ($1.602 \cdot 10^{-19}$ As)

N_d ... donor density (in n-type semiconductor)

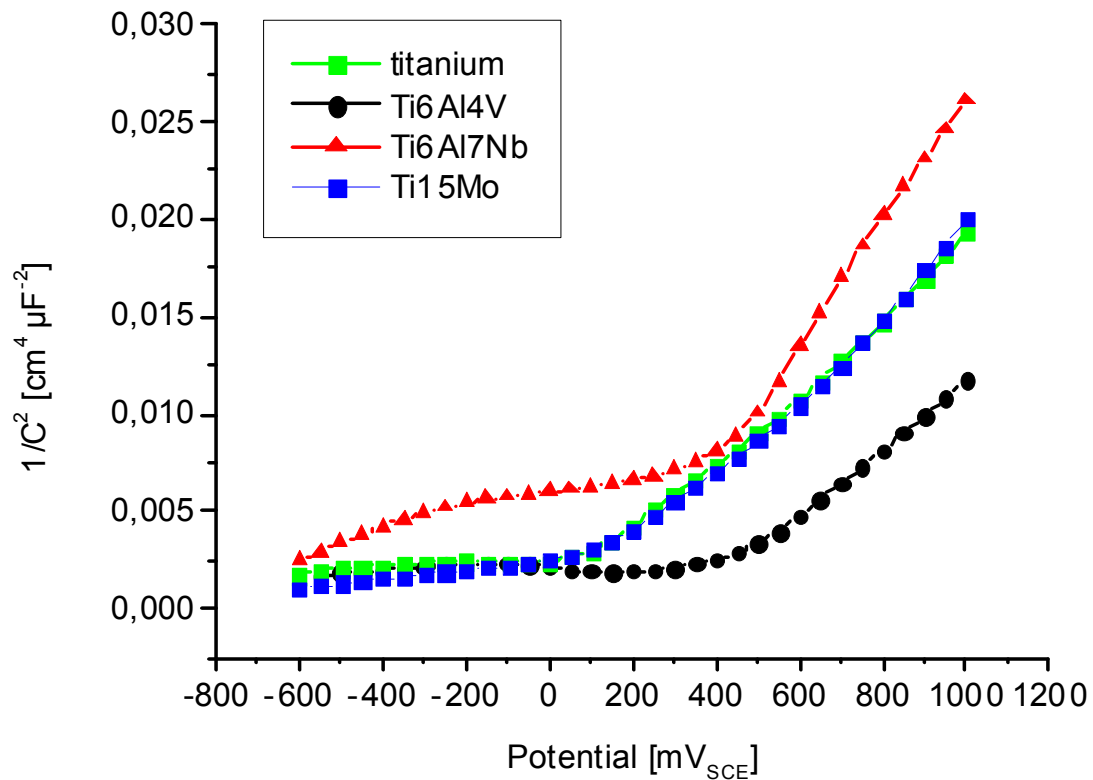
ε₀ ... absolute dielectric constant ($8.86 \cdot 10^{-14}$ F cm⁻¹)

ε ... relative dielectric constant

U_{FB}... flatband potential

k ... Boltzmann constant ($1.38 \cdot 10^{-23}$ Ws K⁻¹)
so with kT/q = 26 mV

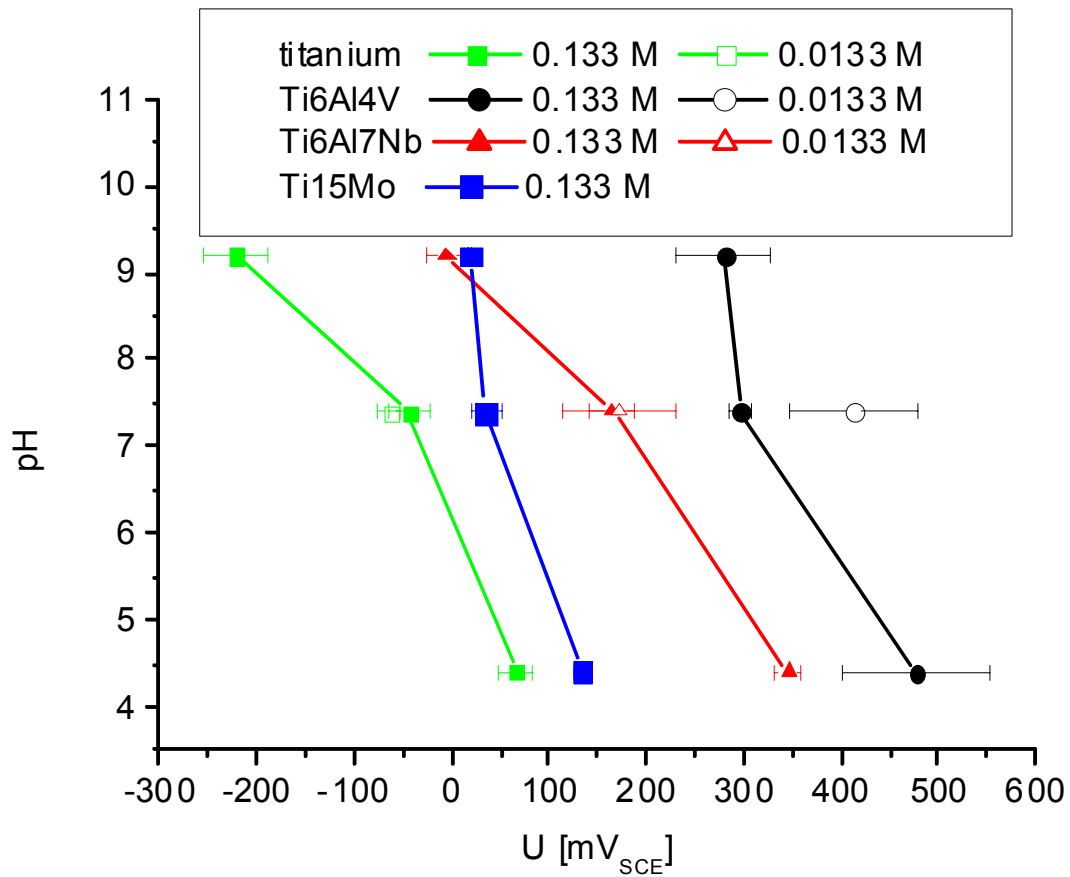
Typical Mott-Schottky-Plots



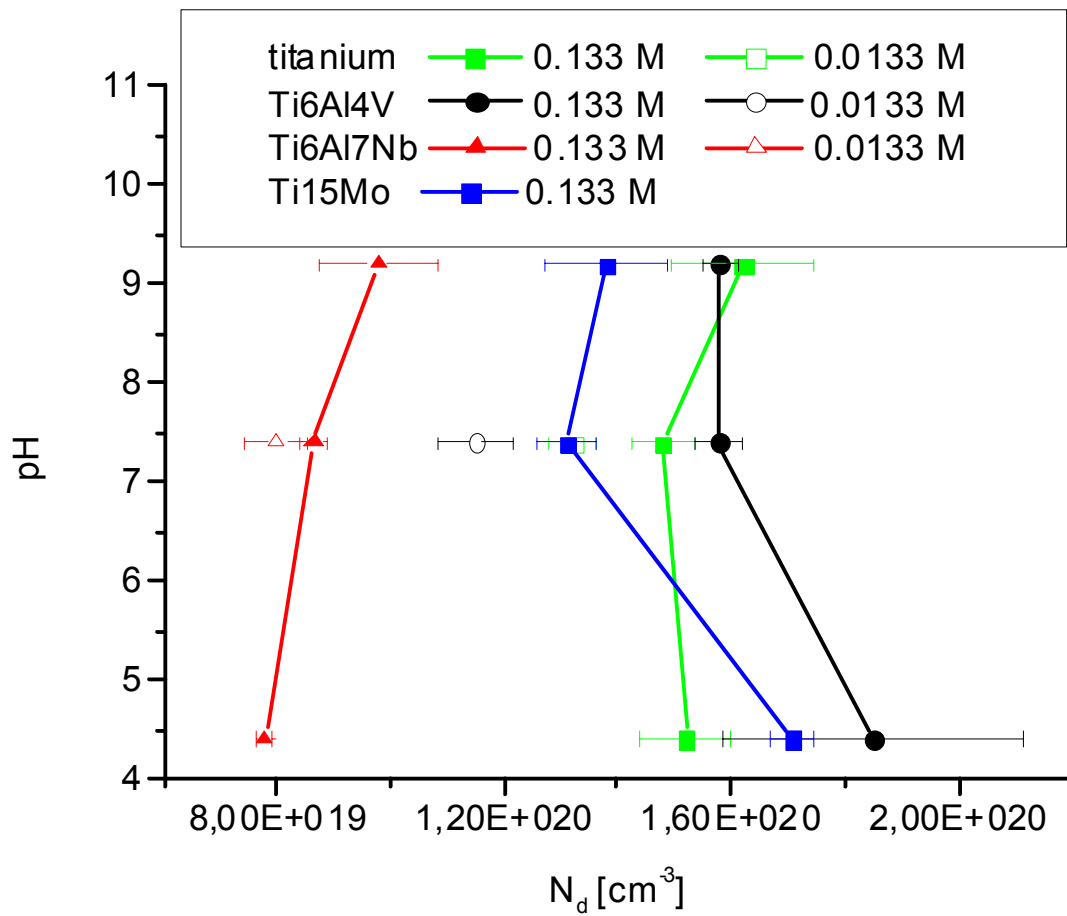
0.133 M Phosphate buffer solution, pH =7.4

typical redox potentials in biological systems 350 – 550 mV_{SCE}
(D. Velten et al., J. Biomed. Mater. Res. **59** (2002) 18.)

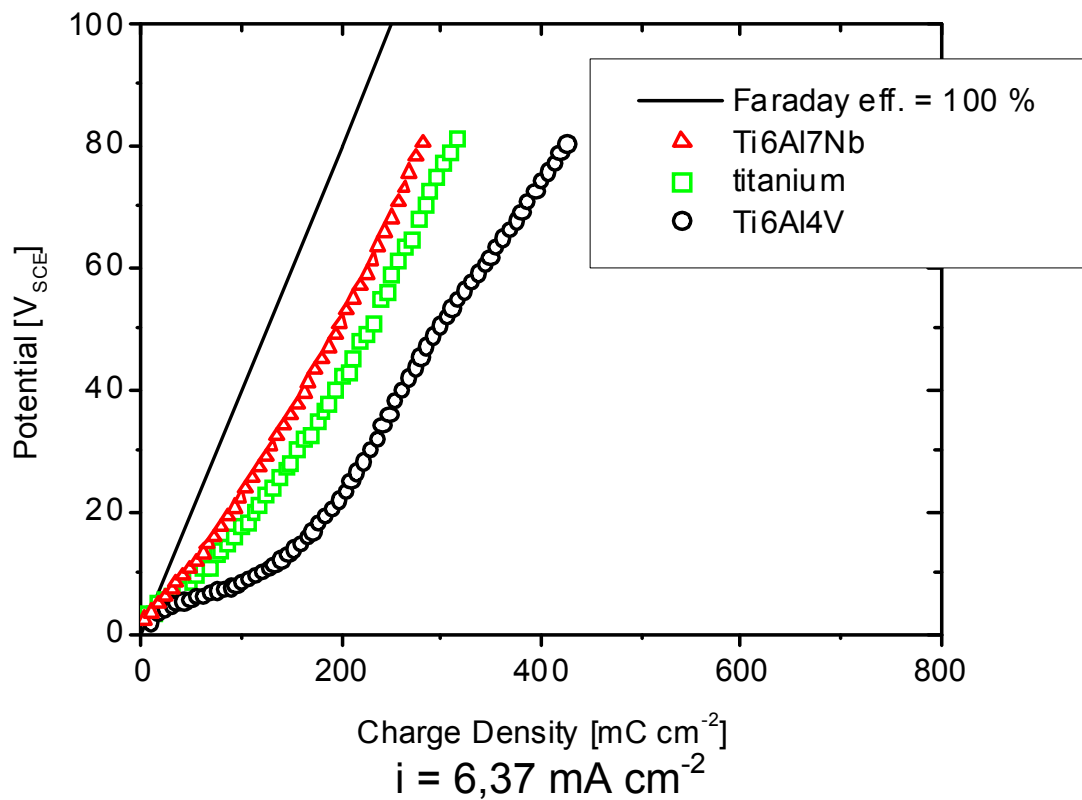
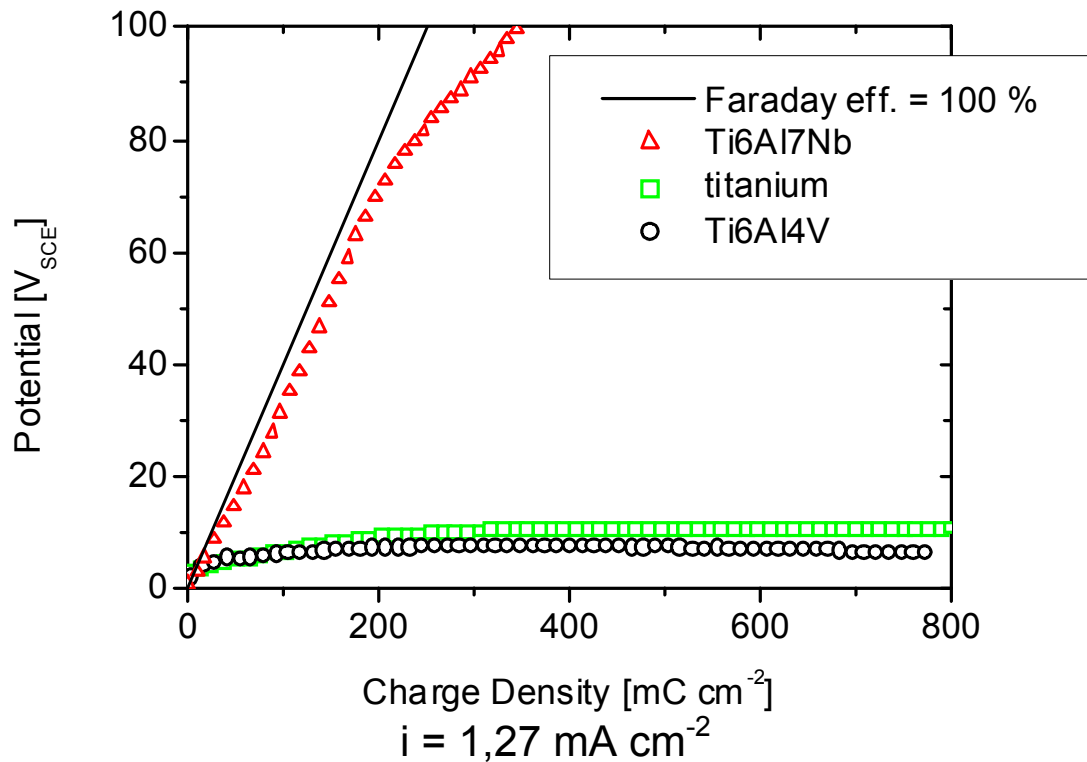
Flatband potentials of different materials as f (pH)



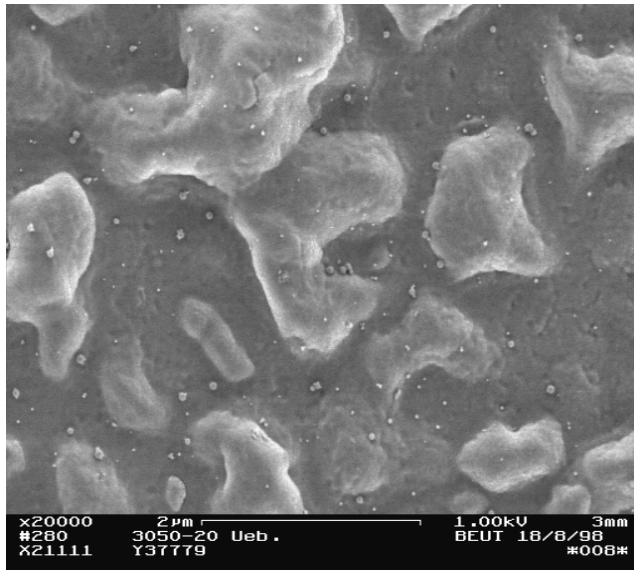
Donor densities of different materials as f (pH)



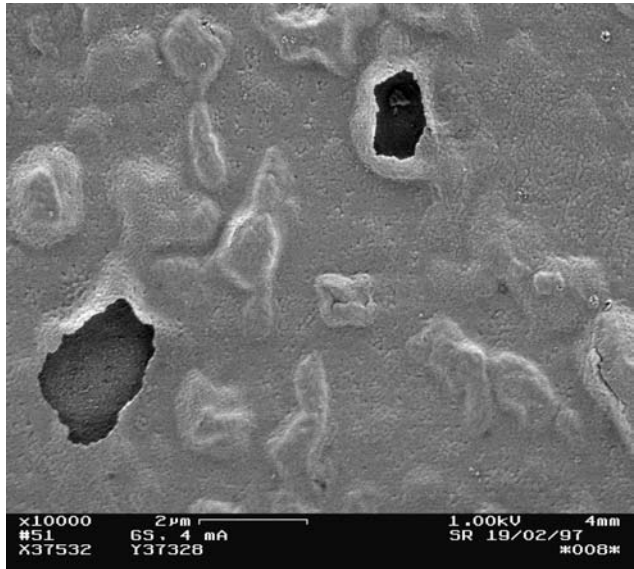
Charge density-potential curves for galvanostatic polarization of different titanium alloys



Morphology of anodic oxide layers



Titanium



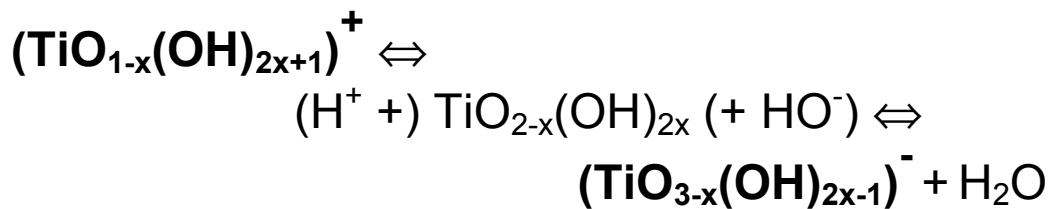
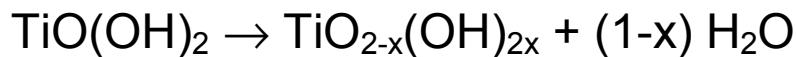
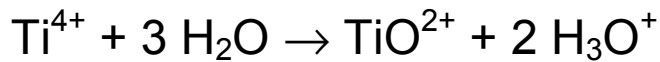
Ti6Al4V



Ti6Al7Nb

Chemical / electrochemical processes

oxide formation in PB oxide/electrolyte

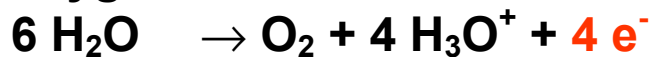


oxide formation in PB metal/oxide

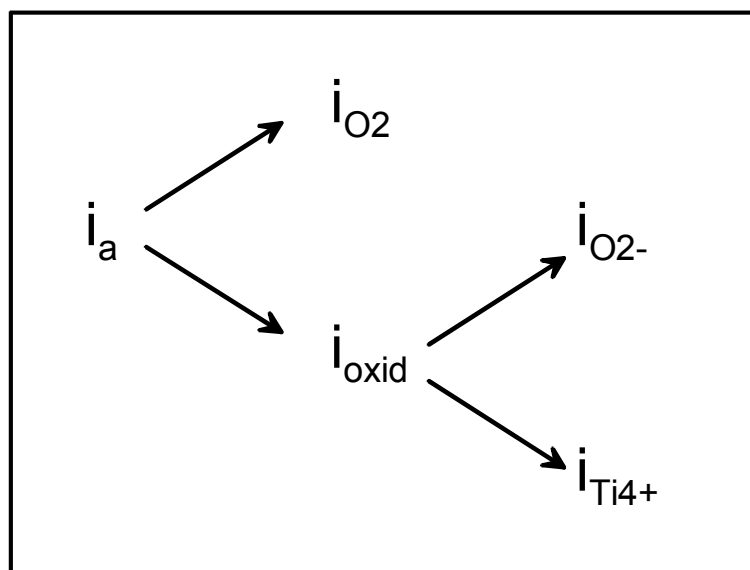


parallel reaction(s)

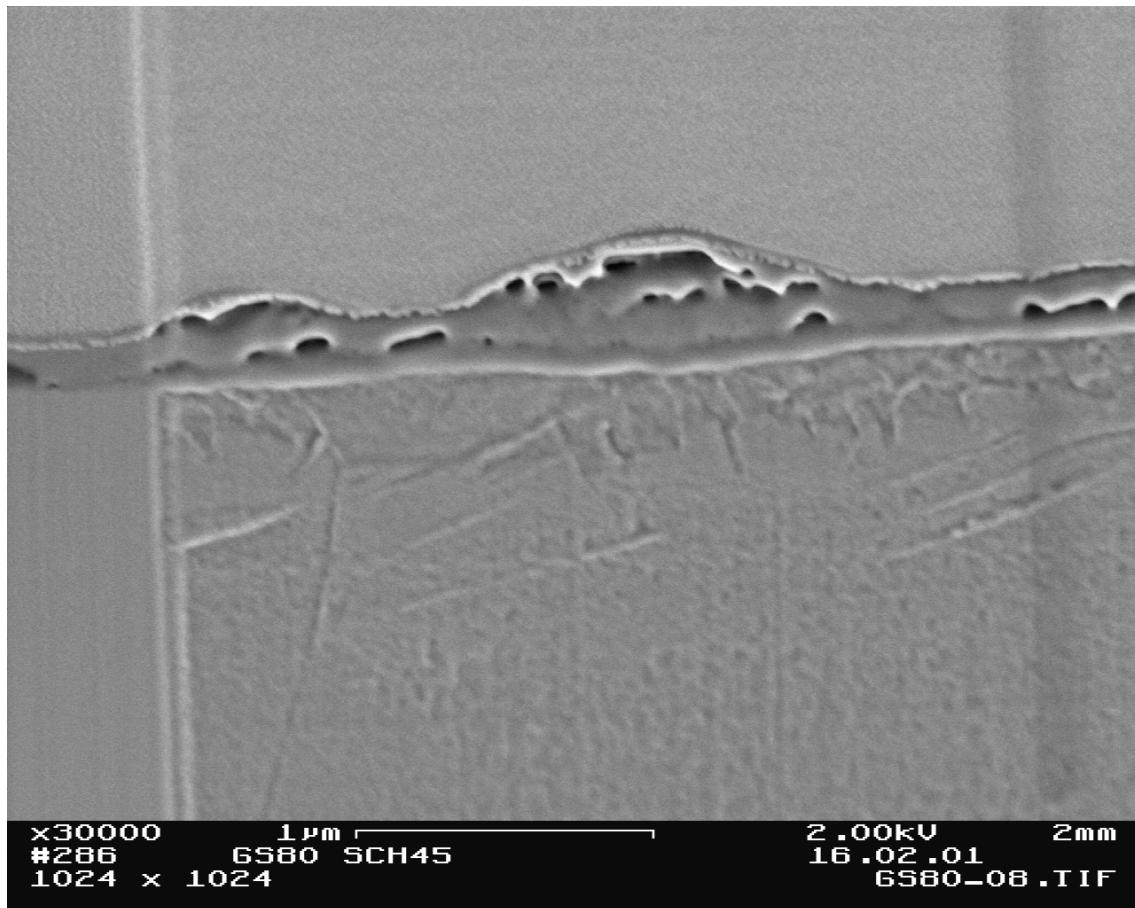
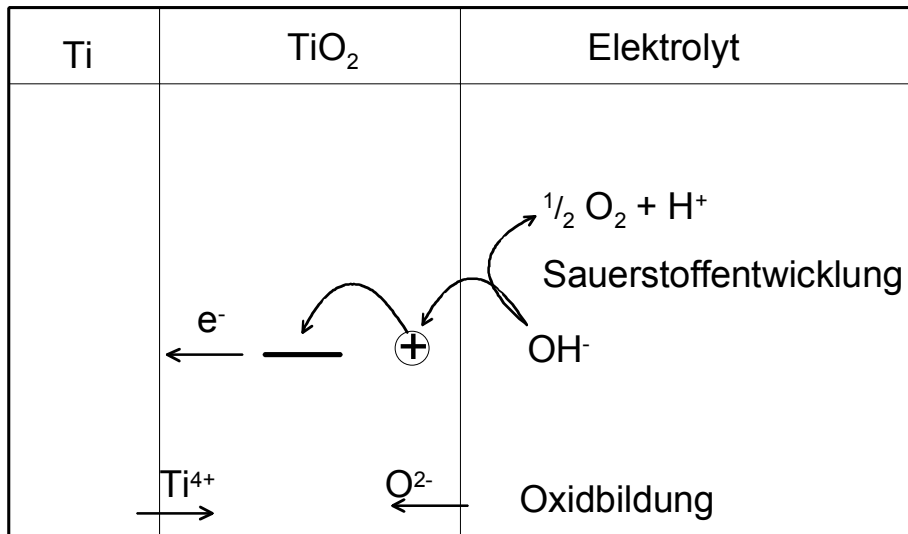
oxygen evolution



redox reactions with adsorbed biomolecules



Oxygen evolution – Where in the oxide layer?



Oxide layer on Ti6Al4V formed with

$$i = 6.37 \text{ mA cm}^{-2} \text{ to } U = 80 \text{ V}_{\text{SCE}}$$

FIB cross section preparation; (top – platinum bar; middle – porous oxide layer; bottom – Ti6Al4V substrate)

8.6. Combinations of metallic biomaterials

Besides pitting corrosion, intergranular corrosion and crevice corrosion, **galvanic corrosion** causes drastic local corrosion, mainly generated by an electron-conducting connection of different materials.

Field of dental surgery - more craftsman-like techniques used in the dental laboratory may cause additional supplementary disadvantageous changes in structure of the materials as result of heat treatment during welding or soldering

Short-circuit potentials and local current investigations of the combining of titanium with several other alloys in a 1 % NaCl solution (pH 6.3), in part with 10 % lactic acid (pH 1.81) showed:

- combination with noble metal alloys: titanium acts as local anode → because of stability of the ion-conducting passive layer **no increased corrosion takes place**
- combination with nickel-based alloys: titanium acts as the local cathode → because of the *n*-semiconducting properties of the titanium passive layer **increased corrosion appears**. Causes **corrosion attack of nickel-based alloys** + in case of unfavourable surface ratios **damage to titanium due to hydrogen embrittlement**.

Using different galvanic couples (Ti/gold-based alloys, Ti/palladium-based alloys and Ti/non-precious alloys), titanium was under either cathodic or anodic control.

It has been found, that alloy is potentially useable for superstructures in galvanic coupling with titanium if:

- in a coupling titanium has a weak anodic polarisation
- the galvanic cell current density is also weak
- the crevice corrosion potential is markedly higher than the coupling potential.

8.7. Special Materials

Goal: development of alloys being able to replace amalgams

→ Gallium based alloys

(typical) compositions:

solid component: 50 % Ag, 25.7 % Sn, 15 % Cu, 0.3 % Pd
60.1 % Ag, 28.05 % Sn, 11.8 % Cu, 0.05 %
Pt

liquid component: 62 % Ga, 25 % In, 13 % Sn, 0.05 % Bi
 F_p 10 °C, (F_p for Gallium 29.8 °C)

Investigations deoxygenated Ringer's solution at 37 °C showed:

Uncoupled gallium-based alloy at low overpotentials:
selective dissolution of divalent tin ions, followed by a dissolution of Ga

Uncoupled gallium-based alloys and gallium alloy coupled with amalgam: anodic current densities 10^3 - 10^4 times higher than that of an uncoupled amalgam → very poor corrosion stability.

Comparison of the cytotoxicity of

- gallium and indium ions
- with that of mercuric ions

in a concentration range of 1 μ M to 1 mM using L-929 mouse fibroblasts:

mercuric ions:

50 % inhibition at concentration of 0.35 mM

gallium and indium ions:

did not significantly inhibit dehydrogenase activity in either growing or the confluent phase

Effects of metal ions used in dental materials on the conversion of amorphous calcium phosphate to hydroxyapatite *in vitro*:

According to effects on both the rate of hydroxyapatite transformation and induction time:

- inhibitory (in the order: Ni, Sn, Co, Mn, Cu, **Ga**, Th, Mo, Cd, Sb, Mg, **Hg**)
- ineffective (Cs, Ti, Cr, Fe^{2+})
- stimulatory (Fe^{3+} , **In**).

Summary

- similar or better mechanical properties than modern (Hg) amalgams
- difficult manipulation
- low corrosion resistance
- lack of gallium's biological impact

→ restricted use